

Research article

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Observing and controlling a Tamm plasmon at the interface with a metasurface

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Abstract: We demonstrate experimentally that Tamm plasmons in the near infrared can be supported by a dielectric mirror interfaced with a metasurface, a discontinuous thin metal film periodically patterned on the sub-wavelength scale. More crucially, not only do Tamm plasmons survive the nanopatterning of the metal film but they also become sensitive to external perturbations as a result. In particular, by depositing a nematic liquid crystal on the outer side of the metasurface, we were able to red shift the spectral position of Tamm plasmon by 35 nm, while electrical switching of the liquid crystal enabled us to tune the wavelength of this notoriously inert excitation within a 10-nm range.

Keywords: Tamm plasmon; metasurface; liquid crystal; Bragg mirror.

1 Introduction

A Tamm plasmon (TP) is a localized resonant optical state, a quasi-particle, which exists at the interface between a metal and a dielectric (or semiconductor) Bragg mirror (BM). It was theoretically predicted by Kaliteevski et al. [1] and experimentally observed by Sasin et al. [2]. The TP

dispersion lies completely within the light cone and, therefore, in contrast to an ordinary surface plasmon polariton, a TP can be excited with both transverse electric (TE)- and transverse magnetic (TM)-polarized light at any angle of incidence [1]. Another advantage of a TP over a surface plasmon polariton is that the former appears to be almost insensitive to dissipative losses in the metal film, as its electromagnetic fields are localized predominantly in the non-absorbing BM [3]. Because of its robust nature, a TP has been regarded as a viable alternative to conventional surface plasmons in a wide range of applications, including optical switches, semiconductor lasers and light emitters, and temperature and refractive-index sensors [4–12]. For many practical applications it is important to realize an external dynamic control over the TP wavelength. Such a task, however, presents a formidable challenge as the fields of a TP reside inside the BM and, therefore, are not accessible from the outside. Correspondingly, the approaches proposed thus far involve the integration of a control element into the very structure of the BM [4, 7, 13–16], which is not always feasible. It has also been shown that the wavelength of a TP could change (although irreversibly) as a result of either patterning the metal film on the microscale [3, 17, 18] or corrugating it on the nanoscale [19].

In this Letter, we report on the first experimental observation of a near-infrared (IR) TP at the interface between a BM and a nanopatterned metal film acting as a non-diffracting optical metasurface. We also found that the discrete framework of the metasurface exposed TP to external perturbations, such as changes of the refractive index in an adjacent medium, which enabled us to dynamically control the wavelength of this weakly coupled optical state in a simple yet efficient way.

2 Results and discussion

Figure 1A presents the design of the structure that was used to observe TPs in our experiments. The structure was based on a silver-coated dielectric BM designed to exhibit

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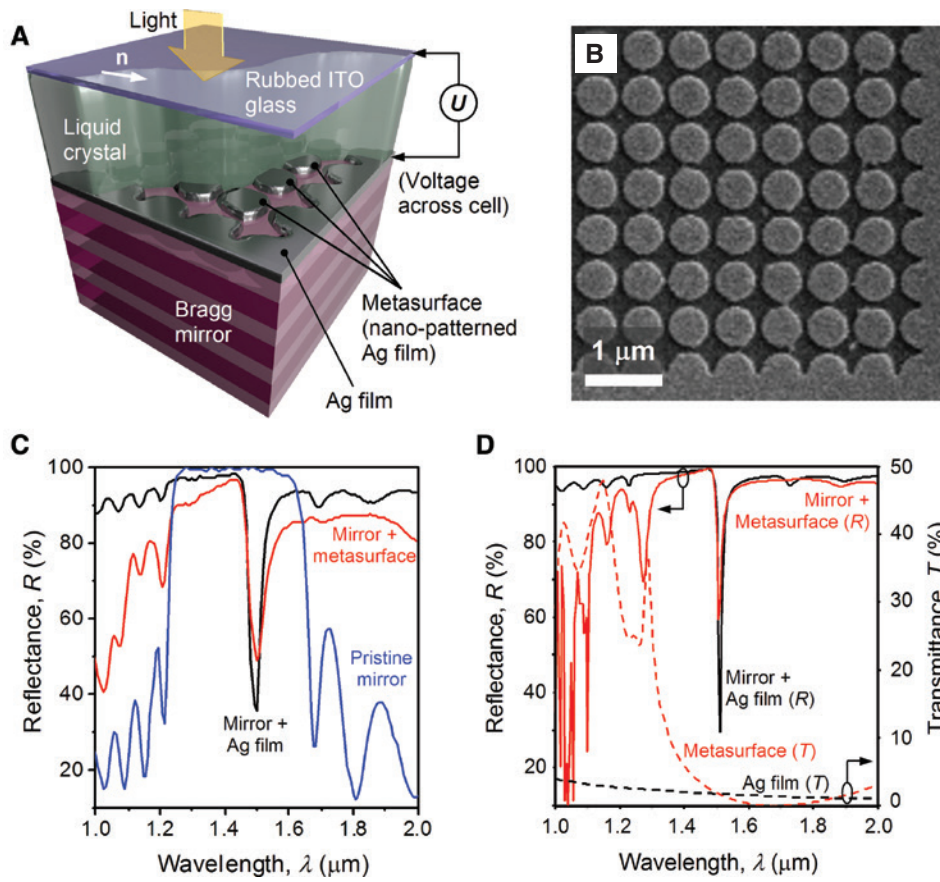


Figure 1: Engineering a Tamm plasmon at the interface with a metasurface.

(A) Schematic of the structure used in our experiments. The white arrow indicates the direction of rubbing (applied to the underside of the ITO cover glass), n , which controlled the alignment of the liquid crystal in the cell. (B) Scanning electron micrograph of a fragment of the metasurface fabricated on top of a Bragg mirror. (C) Experimental reflectivity spectra of the Bragg mirror acquired while it was in a pristine state (blue), after it was interfaced with a 37-nm-thick continuous silver film (black), and after the silver film was nanopatterned to become a metasurface (red). (D) Calculated spectra of the continuous silver film and metasurface. Solid curves show the reflectivity of the film (black) and metasurface (red) placed atop of a Bragg mirror, while dashed curves show the transmission of the film (black) and metasurface (red) residing on a niobium pentoxide substrate.

a 0.5- μm -wide reflection band centered at the wavelength $\lambda = 1.45 \mu\text{m}$. It was formed by a stack of alternating 11 layers of Nb_2O_5 ($n_{\text{Nb}_2\text{O}_5} = 2.24$ [20]) and 10 layers of SiO_2 ($n_{\text{SiO}_2} = 1.47$ [21]). The niobium pentoxide and silicone dioxide layers had the thickness of 159 ± 2 and 246 ± 2 nm, respectively, and were deposited onto a double-sided polished fused silica substrate using magnetron sputtering (PVD 225, Kurt J. Lesker, Jefferson Hills, PA, USA), as detailed by Juškevičius et al. [20]. The silver coating had the thickness of 37 ± 2 nm, and was applied to a section of the BM by magnetron sputtering at room temperature, working pressure of 2.2 mTorr, and deposition rate of about 11 nm/min using an Ag planar sputtering target (99.99% purity, Nova Fabrica Ltd., Ignalina, Lithuania). A $30 \mu\text{m} \times 30 \mu\text{m}$ patch of the silver film was turned into a metallic metasurface by nanopatterning the film with a focused ion beam (Helios Nanolab 600, FEI ThermoFisher Scientific, Hillsboro, OR,

USA). The pattern of the metasurface featured a square array of 550-nm large disks with the period of 600 nm (see Figure 1B), which rendered the nanostructure as non-diffracting above $\lambda = 1.34 \mu\text{m}$, i.e. well within the reflection band of the BM (for normally incident light, the first diffraction order appeared in the adjacent Nb_2O_5 layer at wavelengths shorter than $n_{\text{Nb}_2\text{O}_5} \cdot 0.6 \mu\text{m}$). Such a disk pattern, although very simple, was sufficient to ensure that the array would act as a narrow-band reflector – one of the most basic functionalities of metasurfaces [22]. The reflection band was centered at a wavelength of $\sim 1.65 \mu\text{m}$, as defined by the plasmon resonance of silver nanodisks. The resonant nature of our metasurface implied, in particular, that within the reflection band it would appear optically as dense as the unstructured silver film.

The spectral response of the fabricated sample was characterized in reflection at normal incidence using a

commercial microspectrophotometer (QDI2010, CRAIC Technologies, San Dimas, CA, USA). It employed a cooled near-IR charge-coupled device array with spectral resolution of 0.8 nm and featured a tungsten-halogen light source equipped with a broadband linear polarizer. Light was focused onto the sample (as well as collected) from the metal side using a $\times 15$ reflective objective with numerical aperture 0.28. The reflectivity spectra were acquired through a $22\ \mu\text{m} \times 22\ \mu\text{m}$ square aperture installed in the image plane of the microscope.

Figure 1C compares the reflectivity spectra taken at three different areas of the sample corresponding to an uncoated (i.e. pristine) BM, a BM with a continuous silver film, and a BM with the metasurface. As per design, the pristine mirror is seen to exhibit a characteristic, spectrally flat reflection band spanning from about 1.22 to 1.66 μm (blue curve in Figure 1C). The reflectivity spectrum of the silver-covered area of the mirror reveals the appearance of a narrow reflectivity dip located within the band of the pristine mirror and centered at $\lambda = 1.49\ \mu\text{m}$ (black curve in Figure 1C). Such a conspicuous transformation of the BM's reflectivity spectrum signifies the excitation of the TP, as has previously been shown in a number of works [2, 23–26]. Intriguingly, the BM, when combined with the metasurface, also appeared to support TPs, exhibiting a similar reflectivity dip in the same spectral window (red curve in Figure 1C). While TPs have been shown to exist at the interface with micrometer-sized metal patches [3, 17, 18] and nanocorrugated metal films [19], our results represent, to the best of our knowledge, the first experimental evidence that TPs can also survive the segmentation of metal films into nanometer-sized patches.

Such an outcome is rather remarkable in the view that sub-wavelength metal patches promote very strong coupling of optical fields to leaky modes of an uncovered BM and, therefore, alone cannot support TPs [3]. The key here is to arrange metal patches into an array of sub-wavelength periodicity, which will reduce the leakage to a reasonable level and also ensure that the resulting nanostructure behaves optically as a continuous film (and, thus, can be regarded as a metasurface). Furthermore, within the resonance band of such a metasurface, the polarizability of the metal patches is naturally larger than that of an unpatterned metal film, and hence their scattering cross-section can exceed their physical size [27], effectively making up for the lack of metal between the patches. That, in particular, minimized the distinction between our metasurface and continuous silver film at around 1.5 μm , which was why the nanopatterning did not seem to affect the TP wavelength in the experiment.

Our observations are supported by the results of full-wave numerical modeling (see Figure 1D), which was carried out using commercial simulation software (COMSOL Multiphysics, COMSOL Ltd., Cambridge, UK). Indeed, the modeled reflectivity spectra of the continuous silver film and metasurface placed atop of the BM are seen to exhibit a TP resonance at the same wavelength, $\lambda = 1.51\ \mu\text{m}$. The slight mismatch between the calculated and experimentally measured values of the TP wavelength is attributed to a deviation of the dielectric function of deposited silver from the tabulated data used in our simulations [28]. Note that the transmission of the metasurface alone varies strongly with the wavelength (as dictated by the plasmon resonance), and at $\lambda = 1.51\ \mu\text{m}$ it matches the transmission of a standalone continuous film exactly. The importance of this observation becomes apparent if one recalls that the transmission coefficient of optically thin metal films is almost exclusively controlled by the imaginary part of the refractive index k [29] and, hence, by $\text{Re } \varepsilon$ (as for metals $|k| \gg n$ in the near IR). Given that the reflectivity of the standalone silver film and metasurface at $\lambda = 1.51\ \mu\text{m}$ differs by $< 2\%$ (not shown in Figure 1D), one may expect that their complex dielectric constants at this wavelength will be nearly identical. Indeed, based on the modeled transmission and reflection data, we obtained $\varepsilon = -87 - i12$ and $\varepsilon = -88 - i7$ for the metasurface and silver film, respectively.

While the nanopatterning of the silver film did not affect the ability of the structure to support a TP, it naturally exposed the surface of the BM. Consequently, that should have locally broken the confinement of the TP, allowing a direct access to its fields (which would otherwise remain difficult to couple to, residing under the continuous silver film [8, 30, 31]). To verify that assumption experimentally, we introduced an electrically controlled liquid crystal (LC) cell into the structure of the sample, as schematically shown in Figure 1A. The cell was assembled by placing an indium tin oxide (ITO) cover glass approximately 10 μm above the silver-coated surface of the mirror. The cover glass served as the top (transparent) electrode of the cell, while the silver film played the role of its bottom electrode. The cell was vacuum filled with E7 (Merck KGaA, Darmstadt, Germany), a widely used and commercially available LC mixture with high optical anisotropy ($n_o = 1.50$, $n_e = 1.70$ at $\lambda = 1.55\ \mu\text{m}$ [32]). The surface of the cover glass facing the mirror had been coated with a thin film of uniformly rubbed polyimide to ensure planar alignment of LC molecules in the cell (i.e. parallel to the mirror and along the direction of rubbing, \mathbf{n}). By increasing the voltage across the cell, U , we gradually switched E7 from the planar to homeotropic state in which LC molecules were oriented perpendicular to the mirror. Due to optical anisotropy of LC molecules, the

switching of the cell was accompanied by the change of the LC refractive index from n_e to n_o for light polarized parallel to the direction of rubbing ($\mathbf{E} \parallel \mathbf{n}$). Correspondingly, for light polarized perpendicular to the direction of rubbing ($\mathbf{E} \perp \mathbf{n}$), the refractive index remained n_o .

We note that although the metasurface was electrically discontinuous and, therefore, could not act as the bottom electrode, the section of the cell above it, when inspected under a polarizing microscope, was also displaying the transition between the planar and homeotropic states. Such a behavior was a testament to the long-range nature of the elastic forces arising in nematic LCs. More specifically, the effective range of the elastic forces in an LC cell is comparable to the thickness of the cell [33], which, in our case, ensured that the voltage-induced orientation of LC molecules would be transferred from the periphery of the metasurface to most of the area above it.

Figure 2 presents the reflectivity spectra of the sample integrated with the LC cell, which were measured under linearly polarized light, while sweeping the voltage across the cell from 0 to 2 V. Rather evidently, in the case of unstructured silver film, the spectral location of the TP was unaffected by filling the cell with E7, as well as by changing the state of the LC in the cell (see Figure 2A, B). As noted above, such behavior resulted from strong confinement of the TP fields, which were effectively screened by the unstructured film from the ambient medium and, expectedly, remained inert to external perturbations [8, 30, 31]. By contrast, the TP excited at the interface with the metasurface appeared to be rather sensitive to the presence of the LC (see Figure 2C, D). In particular, filling the cell with E7 red shifted the TP reflectivity dip by about 35 nm ($\mathbf{E} \parallel \mathbf{n}$) and 25 nm ($\mathbf{E} \perp \mathbf{n}$), which was consistent with the increase of the refractive index above the metasurface

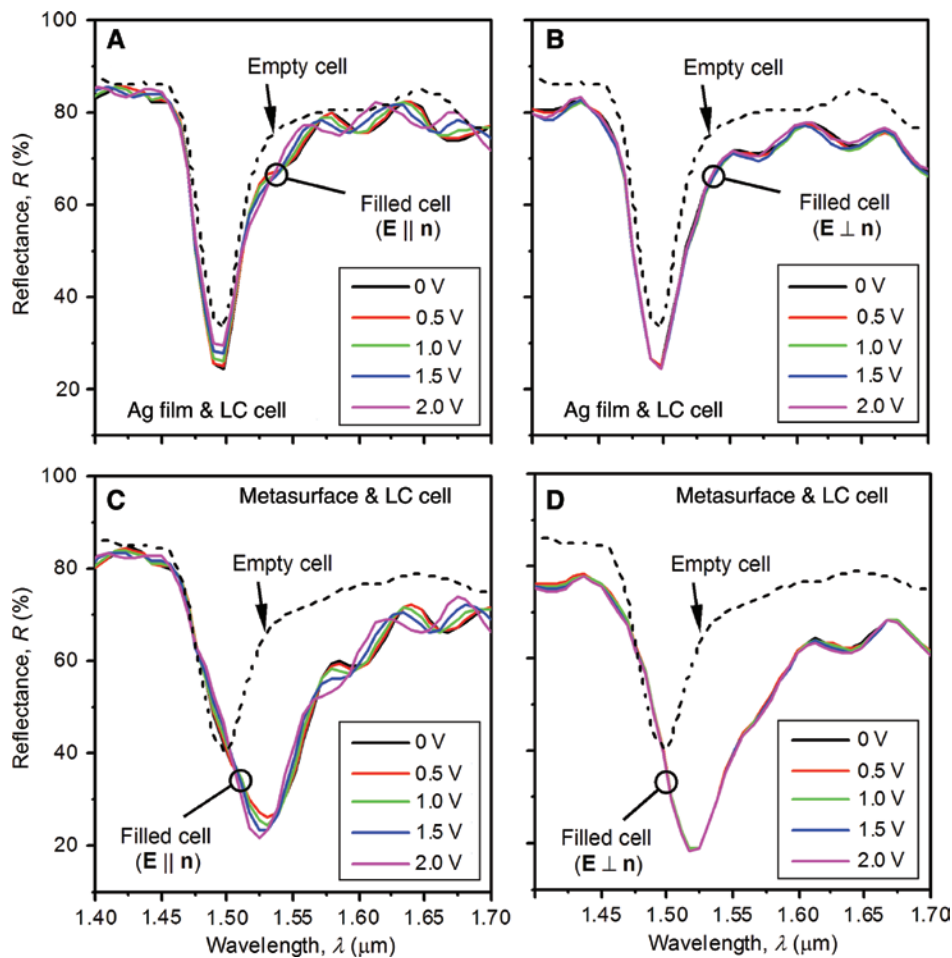


Figure 2: Experimental reflectivity spectra of the silver-coated Bragg mirror acquired with linearly polarized light at different voltages once the structure was integrated with a liquid-crystal cell (solid curves).

Data in panels (A) and (B) correspond to an area of unpatterned silver film, while in panels (C) and (D) – to the metasurface. Dashed curves show the reflectivity spectra of the sample before the cell was filled with the liquid-crystal.

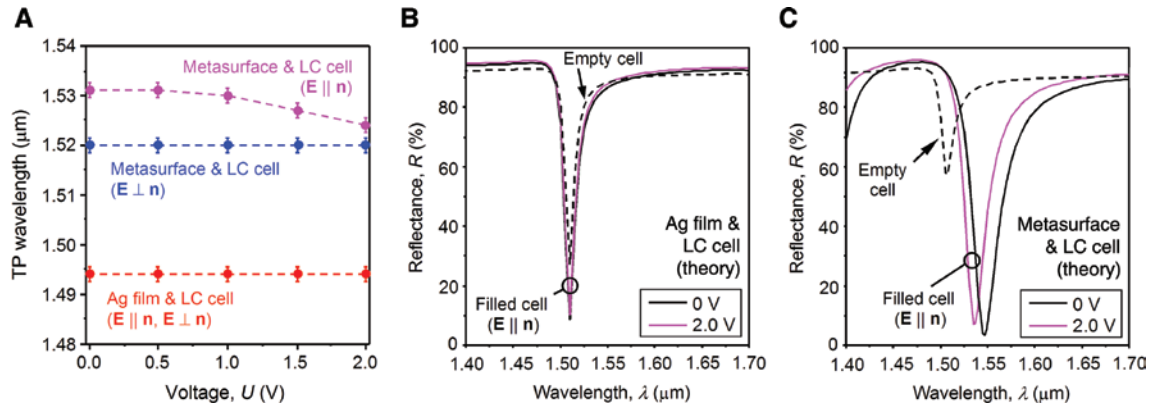


Figure 3: Electrical control of Tamm plasmons (experiment and theory).

(A) Wavelengths of TP resonance plotted as functions of applied voltage. Data points were extracted from Figure 2A–C. Dashed curves are a guide for the eye. Panels (B) and (C) display the calculated reflectivity spectra of a continuous silver film and metasurface atop of a Bragg mirror integrated with an LC cell, respectively. Dashed curves show the reflectivity of the structures before the cell was filled with a liquid crystal. Solid curves correspond to the planar (black) and homeotropic (magenta) states of the liquid crystal.

from 1 to n_e and n_o , respectively. Also, for \mathbf{n} -polarized illumination, the TP reflectivity dip was seen to blue shift as soon as the applied voltage had exceeded 0.5 V, as evident from Figure 2C. The extent of the shift reached 10 nm at $U=2.0$ V (see Figure 3A) and corresponded to the change of the refractive index in the LC cell from n_e and n_o . Our observations, thus, appeared to agree with the assumption we made earlier that the nanopatterning of the silver film would allow external coupling to the fields of the TP. Note that for the orthogonal polarization ($\mathbf{E} \perp \mathbf{n}$) sweeping the voltage did not have any effect on the TP wavelength (see Figures 2D and 3A). Indeed, in that case, the re-orientation of LC molecules occurred in the plane perpendicular to the polarization of light and, therefore, could not affect the relevant component of the refractive index tensor.

The data set presented in Figure 2 also confirms that the spectral shift of the TP reflectivity dip could not result from the selective excitation (due to focused illumination) of TE- and TM-polarized TPs. Indeed, although TP resonances obtained with TE and TM polarizations do not share the same wavelength [1], their excitation would require TE-TM polarization conversion to occur inside the LC and, therefore, could not be exclusive to just one particular case, namely when the metasurface was illuminated with \mathbf{n} -polarized light (Figure 2C). More crucially, we reproduced the observed behavior by modeling in COMSOL the reflectivity of our sample at normal incidence under plane-wave illumination (see Figure 3B, C). Note that the calculated shift of the TP resonance (12 nm) resulting from LC transition between the planar and homeotropic states above the metasurface only marginally exceeds the shift we observed in the experiment, which confirms that the

re-orientation of LC molecules above the metasurface was nearly complete.

Apart from changes in the spectral location of the TP reflectivity dip, we observed consistent, voltage-induced changes of its magnitude under \mathbf{n} -polarized illumination. Interestingly enough, such changes occurred also in the case of unstructured film (see Figure 2A). The exact nature of those changes remains unclear to us. We speculate that the effect might result from mild focusing of light produced by the objective of our microspectrophotometer. Indeed, for light focused on a metal-coated BM, a change in the refractive index immediately above the structure would move its image from the image plane of the instrument and could, in principle, affect the level of the measured reflectivity. Another possible explanation is that a change in the refractive index of the ambient medium would change the reflectivity of its interface with the metal film and, therefore, could modify the conditions for the excitation of a TP. Such an explanation appears to be consistent with the results of our numerical simulations presented in Figure 3B and C, suggesting a different approach to sensing with TPs. A more detailed analysis of the above-noted effect is currently underway and will be published elsewhere.

3 Conclusions

In conclusion, we showed experimentally that a Tamm plasmon could be excited in the near IR at the interface between a dielectric BM and a nanostructured non-diffracting metasurface (which effectively replaced a continuous metal film used conventionally as the second mirror).

Our findings also indicate that the metasurface, through its discrete framework, had enabled an external access to otherwise weakly coupled fields of the Tamm plasmon. More specifically, we found that placing a dielectric, such as a liquid crystal, in direct contact with the outer side of the metasurface red shifted the TP wavelength by as much as 35 nm, while no spectral shift could be detected when the LC was applied to a continuous metal film in the conventional configuration. Furthermore, we managed to tune the TP wavelength within a 10-nm range by changing the LC refractive index above the metasurface with an externally applied electric field. While the reported tuning range is modest compared to what other TP tuning approaches can offer [4, 7, 13–16], we point out that our work aimed at proof-of-principle demonstration of the new mechanism for controlling TPs. There are a number of ways one can improve the efficiency of the new mechanism, which include, for instance, optimization of the patterning and thickness of the metal film, adjusting the composition of the BM, implementing other LC switching modes, and employing LCs with higher optical birefringence (e.g. some of the currently available LC materials exhibit birefringence as high as 0.8 [34]). Hence, we argue that the demonstrated ability to control the spectral location of TP opens up a viable route to exploiting this resonant optical state in many real-life applications, including optical switching, enhancement of optical non-linearity, lasing and light emission, and surface-enhanced spectroscopy.

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